Optical Anisotropy of Polyimide and Polymethacrylate Containing Photocrosslinkable Chalcone Group in the Side Chain under Irradiation of a Linearly Polarized UV Light

Dong Hoon Choi* and Young Kwan Cha

College of Environment & Applied Chemistry, Materials Center for Information Display, Institute of Natural Sciences, Kyung Hee University, I Seocheon, Yongin, Kyungki 449-701, Korea
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Photocrosslinkable soluble polyimide and polymethacrylate compound were synthesized for studying the optically induced anisotropy of the thin films. Chalcone group was introduced into the side chain unit of two polymers. We observed a photodimerization behavior between the double bonds in the chalcone group and an optical anisotropy of these materials by irradiation of a linearly polarized UV light (LPL). Optical anisotropy of the thin film was also investigated by using polarized UV absorption spectroscopy. The dynamic property of optical anisotropy in photoreactive polyimide was compared to that in polymethacrylate containing chalcone group in the side chain.

Keywords: Optical anisotropy, Chalcone, Photodimerization, Polyimide, Polymethacrylate.

Introduction

Polymeric materials have recently attained remarkable interests as photoreactive materials. Photochemical reactions in organic polymeric materials can induce much change in physical properties such as solubility, optical transparency, dielectric constant, and refractive index. A number of studies were performed to fabricate the alignment layer of liquid crystals (LCs) using photoreactive polymers.1-5

The photoalignment technique for LC display has received much attention in recent years because of its clean and photo-patternable system.6-13 Many kinds of photoreactive polymers that can undergo anisotropic isomerization or photocrosslinking reactions have been reported for the photoalignment of LC.8-16 Nematic liquid crystal (NLC) can be aligned homogeneously on the photoreactive polymer film which is usually exposed to a linearly polarized light (LPL).

From the practical point of view, the thermal and photostability of the alignment layer is very important and the alignment layer must be transparent in a visible region for the display device. A photo-crosslinkable polymer system can offer high thermal stability because of its three dimensional network and several photo-crosslinkable polymers for the LC photo-alignment have been investigated.13-16 Polymers containing unsaturated aromatic acid or ester units, especially the cinnamic ester and cinnamic acid derivatives were studied for phototransformation phenomena, which occurred during the irradiation with UV light (λ = 250-300 nm). For example, anisotropic [2+2] cycloaddition of polyvinylcinnamate (PVCi) film can induce irreversible low molecular weight-liquid crystal (LMW-LC) alignment. The polymers containing chalcone group in the side chain were also frequently studied for application to LC photo-alignment

layer. The induction mechanism of optical anisotropy is identical with that in polyvinylcinnamate.17-19 Above polymeric materials could show the possibility to apply them to LC alignment layer only due to angular selective [2+2] photodimerization process. However, most of the photocrosslinkable polymer showed that the optical anisotropy of the thin film increased at the initial stage of UV irradiation and gradually diminished under a continuous irradiation. If the optical anisotropy continue to increase or stabilize under a continuous UV irradiation, LC alignment can be observed to quite stable.

Among many promising photosensitive groups, a chalcone group has been well studied and can be used in photocrosslinkable polymers because it affords high sensitivity to UV radiation (λ = 300-350 nm). Because of poor solubility arising from the rigid-rod nature of chalcone backbone polymers, most research works involved only polymers that contain the chalcone unit in the side chain.20-22

In this work, we also synthesized a new photoreactive soluble polyimide and polymethacrylate. Chalcone group was introduced into the side chain unit of the new photoreactive polymers, which is highly sensitive to the LPL photochemically. The capability of photocrosslinking of new compounds was demonstrated using UV-Vis absorption spectroscopic analysis. The optical anisotropy of the film induced by irradiation of a linearly polarized UV light is also investigated by using the polarized UV absorption spectroscopy. Photocrosslinking was performed to elaborate the very dense and rigid matrix after UV exposure and angular selective photochemical reaction was performed to induce high optical anisotropy.

Experimental Section

Scheme 1 and 2 show the synthetic schemes of the
photosensitive material such as soluble polyimide and polymethacrylate used in this study.

**Synthesis of 4-hydroxychalcone**

(3-(4-Hydroxy-phenyl)-1-phenyl-propenone): To a solution of 10 g (0.082 mol) of 4-hydroxybenzaldehyde in 300 ml ethanol, 29.52 g (0.246 mol) acetophenone was added and the reaction mixture was cooled, followed by the addition of 6.56 g (0.164 mol) of sodium hydroxide dissolved in 11.66 mL of water. The reaction mixture was stirred for two days at room-temperature. After evaporation of ethanol under reduced pressure, the product was dissolved in ethylacetate and washed with water. The solution was dried under Na2SO4 and concentrated. Then precipitation in ethylacetate and washed with water. The solution was done in cold hexane to give 70% yield. m.p. 178 °C.

1H-NMR (DMSO-d6) δ (ppm) 7.5 (1H in ethylenic -CH=CH-C(=O)-), 7.85 (1H in ethylenic -CH=CH-C(=O)-), 6.9-8.1 (9H in aromatic group), 7.4-8.0 (9H in aromatic, m).

**Synthesis of 4-(2-hydroxyethoxy)chalcone:** To a solution of 5 g (0.0223 mol) of 4-hydroxychalcone in 50 mL of ethanol, 1.784 g (0.045 mol) NaOH dissolved in 3.17 mL of H2O was added and the reaction mixture was cooled, followed by the addition of 5.4 g (0.067 mol) of 2-chloro-ethanol. The reaction mixture was stirred for 24 hr at 50 °C. After evaporation of ethanol under reduced pressure, the product was dissolved in CH2Cl2, washed with 1N NaOH aqueous solution and then purified by crystallization from CH2Cl2/petroleum ether (60 mL/80 mL) to give 62% yield. m.p. 52 °C.

1H-NMR (DMSO-d6) δ (ppm) 4.05, 4.15 (2H, 2H in O-CH2-CH2-O-), 7.45 (1H in ethylenic -CH=CH-C(=O)-), 7.8 (1H in ethylenic -CH=CH-C(=O)-), 6.9-8.1 (9H in aromatic, m).

**Synthesis of photoreactive polymethacrylate (PMAC):** To a solution of 4 g (0.0178 mol) of 4-hydroxychalcone in 100 mL benzene, 2.17 g (0.0267 mol) triethylamine was added. 2.8 g (0.0267 mol) of methacryloyl chloride was added dropwise into the solution at room temperature. The reaction mixture was stirred for 24 hr. After washing the mixture with water three times, it was then concentrated to a half volume of the initial one. It was poured into hexane to obtain the precipitate. We could collect the white solid methacrylate product (yield, 82%).

1 g (0.0034 mol) of the methacrylate monomer was dissolved in 15 mL of NMP and 0.0148 g (0.09 mmol) of AIBN were also added to the reaction mixture. The reaction mixture was stirred for two days at 65-70 °C under argon atmosphere. After precipitation into methanol, the solid product was purified by precipitation from methanol/THF to give 90% yield. Tg 103 °C.

1H-NMR (DMSO-d6) δ (ppm) 1.4-1.6 (3H in CH2-(CO-)O C(CH2)-), 1.6-1.7 (2H in CH3-(CO)-O C(CH2)-), 7.5-7.6 (1H in ethylenic -CH=CH-C(=O)-), 7.7-7.8 (1H in ethylenic -CH=CH-C(=O)-), 7.4-8.0 (9H in aromatic group)

**Synthesis of chalcone-based soluble polyimide (CSPI):** 1 g (2.25 mmol) of 4,4′-(hexafluoroisopropylidenediethyl phalic anhydride was added to a solution of 0.824 g (2.25 mmol) of 2,2-bis(3-amino-4-hydroxyphenyl)hexafluoropheno- pane) in 20 mL, NMP.

After the reaction mixture was stirred for 12 hrs at room temperature, 0.53 g (6.75 mmol) of pyridine and 0.92 g (9 mmol) of acetic anhydride were added dropwise followed by stirring for 1hr at room temperature. It kept stirring for 4hrs at 50 °C. After filtration, the product (soluble polyimide, SPI) was obtained by precipitation into methanol to give 70% yield.

To a solution of 1 g (1.3 mmol) of SPI in 10 mL THF, 1.04 g of (3.87 mmol) 4-(2-hydroxyethoxy)chalcone and 1.705 g (6.5 mmol) of triphenylphosphine (PPh3) were added and the reaction mixture was cooled to 0 °C, followed by the addition of 1.314 g (6.5 mmol) of disopropyl azodicarboxylate (DIAD). The reaction mixture was stirred for 12 hr at 80 °C. After filtration, the product (CSPI) was obtained by precipitation into methanol to give 65% yield. Tg 158 °C.

1H-NMR (DMSO-d6) δ (ppm) 4.2-4.5 (8H in -O-CH2-, 7.4-7.5 (2H in ethylenic -CH=CH-C(=O)-), 7.8-7.9 (2H in ethylenic -CH=CH-C(=O)-), 7.2-8.2 (30H in aromatic group)

**Characterization**

UV-Vis absorption spectroscopy: Absorption spectra were obtained using a Hewlett Packard UV-Vis 8453 spectrophotometer (PDA type, λ = 190-1100 nm). The polymers were dissolved in THF and cast onto quartz plate followed by drying in vacuo at 50 °C for 2 hr. All the polymer films on quartz plates were irradiated with a linearly polarized He-Cd laser (Liconix 3675NX, λ = 325 nm, Imax = 210 mW/cm2). The intensity was measured by using a 13PEM001 (MELLES GRIOT) detector and power meter.

For the polarized absorption spectroscopic study to investigate the dichroism and the order parameter during UV illumination, we equipped Glan-Taylor polarizer in front of the probe beam in the spectrophotometer.

FT-IR spectroscopy: Film specimens of the synthesized polymers on KBr window were prepared from tetrahydrofuran solution (2%). After the solvent was evaporated, the films were stored under vacuum at 50 °C for at least 3 hours to remove the residual solvent completely. The sample on the KBr window was subjected to random polarized UV light irradiation and the FT-IR spectra were taken with Bruker IFS66v before and after UV exposure.

NMR spectroscopy: Proton NMR spectrum was recorded with Varian 200 NMR spectrometer. DMSO-d6 was used as a solvent for recording NMR spectra. Degree of substitution of chalcone group in polyimide (CSPI) was determined by using peak integration method.

**Results and Discussion**

Synthesis and characterization: Synthetic procedures were simply illustrated in Scheme 1 and 2. Photoreactive polyimide and polymethacrylate contain unsaturated carbon-yloieties as photosensitive groups in the side chain. 4-Hydroxychalcone and 4-(2-hydroxyethoxy)chalcone were synthesized following the conventional method, which gave a good yield. In polymethacrylate (PMAC), we firstly...
synthesized the methacrylate monomer using methacryloyl chloride and 4-hydroxychalcone. The methacrylate monomers were polymerized under a radical initiator. The soluble polyimide was synthesized by condensation of specific di-anhydride and diamine bearing two trifluoromethyl groups, respectively. In this case, the diamine has two hydroxy groups in the benzene ring. In order to avoid the geometrical hindrance and lowering the reaction yield, we anchored the chalcone with ethylene spacer such as 4-(2-hydroxyethoxy)-chalcone. The substitution was successfully performed and the degree of substitution was found to be 80% determined by NMR spectroscopic technique. The structures of two polymers were well confirmed by NMR spectral analysis. (See Figure 1). The intrinsic viscosities of PMAC and CSPI were determined to be 0.07 and 0.20 in tetrahydrofuran at 25 °C, respectively. Those polymers are well soluble even in acetone, chloroform, methylene chloride, tetrahydrofuran etc. The glass transition temperatures, T_g, of PMAC and CSPI is 103 and 158 °C, respectively. (See Figure 2)

UV-Vis absorption spectroscopic study in photocross-
linking reaction: The photoirradiation of the polymer films was carried out by using a He-Cd laser (I = 210 mW/cm²). The UV-Vis absorption spectra of the thin film of polyimide (A) and polymethacrylate (B) are given in Figure 3. The spectra were recorded during UV irradiation simultaneously.

The UV absorption changes at \( \lambda_{\text{max}} \) in the two compounds caused by photoreaction were investigated using thin films on quartz plate. Chalcone units underwent a crosslinking reaction by [2+2] cycloaddition between a UV excited chalcone group and an unexcited chalcone group (ground state) on another. The disappearance of the double bond in chalcone unit was observed clearly in this spectroscopic experiment. The maximum absorbance due to the \( \pi \)-electronic conjugation decreased rapidly. In Figure 4, the change of the absorbance was fitted to the stretched exponential decay function to determine the rate constant, k in photodimerization. The time constants, \( \tau (=1/k) \) of polyimide and polymethacrylate were determined 42.86 and 11.8 sec, respectively.

\[
A(t)/A(0) = \exp \left[ -(kt) \beta \right] + R = \exp \left[ -(t/\tau) \beta \right] + R \quad (1)
\]

Polyimide shows a relatively small reaction rate compared to that of polymethacrylate, which is attributable to the rigidity of the main chain backbone. Additionally, chalcone groups are geometrically restricted to retard the photodimerization. It can be conjectured that the stabilized cyclobutane structure could not be formed by intramolecular reaction but by intermolecular reaction. The density of the chalcone group in PMAC is higher so that intermolecular crosslinking reaction can be more favorable. Resulting from above reasons, the residual absorbance (R) of CSPI is larger than that of PMAC.

The infrared spectroscopy of polyimide and polymethacrylate before and after UV exposure: In order to monitor the photoreaction of the olefinic, -C=C- double bonds of the chalcone group, we utilized the FT-IR spectroscopy. Figure 5 shows the localized FT-IR spectra of the polyimide (A) and polymethacrylate (B). The spectra, plotted on an absolute absorbance scale in the range from 1900 to 800 cm\(^{-1}\), were recorded from the same sample before and after UV exposure (light source: high pressure Hg lamp with 365 nm bandpass filter, \( t = 30 \) min, \( I = 23 \) mW/cm²). There are significant changes in the infrared spectra of two compounds indeed occurred upon the UV-irradiation. The -C=C-stretching vibration mode at 1601-1602 cm\(^{-1}\) is observed to decrease significantly by UV exposure. Compared to the absorbance change at 1601 cm\(^{-1}\) in PMAC, the extent of decrease of the absorbance at 1602 cm\(^{-1}\) is relatively small. We can expect two factors for this phenomenon. First, it can be due to the difference of the rigidity of the backbone and the density of chalcone group. Additionally, this is attributed to the fact that the density of the double bond in the benzene ring of polyimide is much higher than that of PMAC. The absorption spectral band from ethylenic double bond is normally superposed with the absorption bands of the double bond in the benzene ring. In addition, the intensity of the C=O unsaturated ketone carbonyl stretching band at 1663-1667 cm\(^{-1}\) decreases systematically with UV exposure time and a new absorption band, attributed to a saturated ketone carbonyl stretching vibration, at 1681-1682 cm\(^{-1}\) appears in the spectra of the two polymers after UV irradiation.

Polarized UV-Vis absorption spectroscopy under irradiation of a linearly polarized light: The films of the two
polymers showed anisotropic spectra after exposure to LPL. We recorded the absorption spectrum every 0.5 second during irradiation of a linearly polarized He-Cd laser (I = 210 mW/cm²). In Figure 6, the representative polarized absorption spectra were illustrated using the two polymers. We can observe the change in UV absorption spectrum and dichroism. $A_{||}$ denotes the parallel absorption and $A_{\perp}$ perpendicular to electric field of the incident LP-UV pump light. Exposure time dependence of the parallel and perpendicular absorbance at $\lambda_{\text{max}}$ was investigated during irradiation of the linearly polarized UV light from He-Cd laser. The decrease in the absorption at $\lambda_{\text{max}}$ was evident, which is attributed to the formation of cyclobutane rings through [2+2] cycloaddition of the carbon-carbon double bonds in chalcone moieties. The parallel absorbance at $\lambda_{\text{max}}$ was observed to be smaller
than the perpendicular absorbance in the whole range of the irradiation time period. Using the two traces of the absorbance, we could analyze the optical anisotropy of the thin film in terms of the difference of the absorbance and the dichroism.

In Figure 7, the difference spectra of the absorbance between the perpendicular and parallel to the polarization direction of pump beam were shown. The difference of the absorbance ($\Delta A = A_\parallel - A_\perp$) of polyimide and polymethacrylate at $\lambda_{\text{max}}$ reached to the minimum after 70 and 10 sec exposure, respectively. Then the difference value increased gradually under further UV irradiation. Therefore, for longer period of UV exposure, the anisotropy of the film was diminished due to the formation of a higher extent of cyclobutane in any directions that is far away of polarization direction of UV light.

Comparison of the optical anisotropy using the polyimide and polymethacrylate: In order to investigate the UV-induced optical anisotropy precisely, we observed the absorbance change during irradiation of a linearly polarized He-Cd laser simultaneously as a function of irradiation time. The absorption spectra were recorded by every 0.5 sec under irradiation of the probe beam whose polarization is parallel and perpendicular to the polarization direction of the pump beam. In Figure 8, we could observe the traces of the parallel and perpendicular absorbance in the whole range of the irradiation time period. Using the two traces of the absorbance, we could analyze the optical anisotropy of the thin film in terms of the difference of the absorbance and the dichroism.

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We employed the dichroism ($\Delta A = A_\parallel - A_\perp$) to compare those two anisotropic behaviors precisely. The dichroism could be calculated by subtracting the perpendicular absorbance, $A_\perp$, from parallel absorbance, $A_\parallel$, which describes the optical anisotropy in the film. Exposure time dependence of the dichroism was shown in Figure 9. In the case of polymethacrylate, the dichroism decreased to -0.185 after 7.8 sec exposure. (See Figure 9B) Then, the value gradually increased to -0.09 until 300 sec exposure. Therefore, the optical anisotropy from the uniaxially aligned reactive molecules was reduced by contributing the photochemical reaction of the molecules away of the polarization direction of the pump beam.

In the case of the polyimide in the initial stage of irradiation, the dichroism decreased generally faster. After 20 sec exposure, the dichroism reached to -0.095 only that is far high. After long time exposure (300 sec), the value of dichroism increased to -0.043. Two behaviors result in the difference of optical anisotropy under a LPL. The poly-

Figure 8. Change of polarized absorbance at $\lambda_{\text{max}}$ as a function of UV exposure time. A: CSPI, B: PMAC.

Figure 9. Exposure time dependence of dichroism of the two polymer films. A: CSPI, B: PMAC.
Optical Anisotropy of Polyimide and Polymethacrylate

The order parameter was calculated using the following equation described to observe the inducing optical anisotropy. The exposure time dependence of the order parameter was parallel and perpendicular absorption spectra. In Figure 10, methacrylate with lower $T_g$ showed much higher optical anisotropy than the polyimide. As we mentioned previously, the density of the chalcone group in polymethacrylate is higher than that of polyimide. Since two chalcone units are tethered in the diamine unit of this polyimide, the distance between two groups is longer. Also, the backbone in this polyimide is much more rigid than that in the polymethacrylate.

We also could calculate the order parameters using the parallel and perpendicular absorption spectra. In Figure 10, exposure time dependence of the order parameter was described to observe the inducing optical anisotropy. The order parameter was calculated using the following equation (2).

$$\text{Order Parameter (O.P.) = } \frac{[A_\parallel - A_\perp]}{[2A_\parallel + A_\perp]} \tag{2}$$

In the sample of polyimide, the order parameter decreased to -0.095 after 200 sec exposure. Then, the value increased in a small positive rate during continuous irradiation due to reduction of the optical anisotropy. Contrary to that phenomenon, the value from the sample of polymethacrylate decreased quite fast at the initial stage of irradiation. And then the following reaction was retarded slightly.

Assuming that the order parameter and its changing behavior are mainly attributed to density of chalcone group and the geometrical hindrance around chalcone groups, this behavior in polymethacrylate is quite reasonably recognized. Although the irradiation continued longer subsequently, the order parameter of the sample of polymethacrylate decreased continuously to show higher anisotropy compared to that in the sample of polyimide. This indicates that the azimuthal dimerization between chalcone groups is hardly formed in polymethacrylate film, since the rate of photodimerization is very large so that the conversion of cyclobutane may be very high. There is a very low concentration of the isolated chalcone group in polymethacrylate after 200 sec irradiation of UV light.

The optical anisotropy from the linearly organized molecular structure in polymethacrylate and polyimide can be investigated and comparison was performed under our strategy. The properties of liquid crystal cell fabricated with the alignment layer of two polymers will be described in our future report.

Conclusion

We synthesized photo-crosslinkable polyimide and polymethacrylate comprising chalcone group. Cyclobutane structures were formed under UV irradiation in two compounds between the side chains that were confirmed by spectroscopic analysis.

We clarified anisotropic LP-UV photoreaction of polyimide and polymethacrylate compound. The dichroism of the photoreacted chalcone group was negative and very large when using as-coated polymethacrylate films. Owing to a large rate of photodimerization between the chalcone groups and the lower rigidity of the main chain, the value of optical anisotropy from the photodimerization was sustained much longer even under long-term UV exposure.

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